

Manipulating electronic structure of nickel phosphide via asymmetric coordination interaction for anion-exchange membrane based seawater electrolysis



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ARTICLE INFO

Keywords:

Asymmetric construction
Bifunctional electrocatalyst
Seawater splitting
Anion-exchange membrane water electrolyzer

ABSTRACT

Alkaline seawater electrolysis faces serious problems such as low catalytic activity of bifunctional catalysts and vulnerability to Cl^- corrosion. Hence, Ru-doped nickel phosphide asymmetric structure ($\text{Ru-Ni}_2\text{P}/\text{Ni}_5\text{P}_4$) has been synthesized as highly efficient and corrosion resistant electrocatalysts for alkaline seawater electrolysis at high current density. In-situ characterization technique and DFT calculations reveal that compared with the monophase nickel phosphide, the construction of Ni_xP asymmetric structure manipulate the electronic structure and d-band center, promoting the adsorption of intermediates, as well as accelerate phase transition of the OER process. Moreover, the weak adsorption energy of Cl^- on the surface of the $\text{Ru-Ni}_2\text{P}/\text{Ni}_5\text{P}_4$ catalyst avoids the corrosion of Cl^- in seawater. Consequently, $\text{Ru-Ni}_2\text{P}/\text{Ni}_5\text{P}_4$ achieve 1000 mA cm^{-2} with overpotentials of only 128 and 450 mV for HER and OER, respectively. After assembling the Anion-exchange-membrane water electrolyzer (AEMWE), $\text{Ru-Ni}_2\text{P}/\text{Ni}_5\text{P}_4$ shows high average efficiency and economic value at different current densities, and demonstrates its corrosion resistance to Cl^- and potential for industrial applications.

1. Introduction

Hydrogen, as an important product of converting renewable energy into clean energy, is an important part of achieving the dual-carbon goal and building a clean energy system [1–3]. Electrochemical water splitting is a safe and efficient way to produce hydrogen. Nevertheless, seawater resources, which are more abundant than pure water resources, have attracted more widespread attention [4–6]. However, seawater contains a large amount of Cl^- , and the thermodynamic similarity between the oxygen evolution reaction (OER) and the chlorine oxidation reaction (ClOR) leads to the production of ClO^- , which severely corrodes the catalyst, weakening its activity and durability [7–9]. Currently, noble metal catalysts such as RuO_2 and Pt/C have exhibited high efficiencies in both OER and hydrogen evolution

reactions (HER), however, its large-scale application is limited by scarcity and high cost [9,10]. Therefore, achieving high current densities at low voltages while developing bifunctional catalysts is a top priority [11].

In recent years, transition metal phosphides (TMPs) have attracted much attention as bifunctional catalysts, and it is noteworthy that due to the high electronegativity of the P atoms in the TMPs, they can be used as a moiety to capture positively charged protons, thus providing high activity for the desorption of H_2 [12,13]. In contrast to the traditional symmetric structure, the electronic structure of the catalysts can be modulated by rationally constructing the asymmetric coordination structure [14–16]. Wang et al. [17] prepared asymmetric Janus heterostructure of $\text{CeO}_2/\text{ZnCoS}$ ($\text{J-CeO}_2/\text{ZCS}$) through a heterogeneous interfacial induction strategy, which owns the ability to efficiently

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regulate the electronic structure and charge redistribution, thereby enhancing the redox activity of the active site and the uptake/desorption of reaction intermediates. Therefore, utilizing the asymmetric coordination to modulate the electronic structure of phosphides at the interface may be an effective strategy to facilitate the seawater electrolysis [18–20]. For HER, the H^* adsorption strength is considered to be an intuitive indicator for evaluating the performance of HER, and the electron cloud density near the d-orbitals as well as the position of the d-band center affect the HER performance of the catalyst [21–24]. For OER, by adjusting the charge distribution at the phosphides interface, a phase transition can occur at low voltage, thus serving as a true active site to accelerate the OER kinetics [25,26]. However, utilizing the asymmetric coordination to manipulate electronic structure of phosphides as bifunctional electrocatalysts for seawater electrolysis is barely reported and the reaction mechanisms and dynamic evolution of the

HER and OER, particularly concerning the asymmetric coordination to enhance the corrosion-resistance of seawater electrolysis is even less reported [27,28].

Based on the above-mentioned, Ru-doped nickel phosphide asymmetric structure ($\text{Ru-Ni}_2\text{P}/\text{Ni}_5\text{P}_4$) has been synthesized, exhibiting an excellent bifunctional performance for alkaline seawater electrolysis. In-situ characterization technique and DFT calculations reveal that compared with the monophase nickel phosphide, the construction of Ni_xP asymmetric structure manipulate the electronic structure and d-band center, promoting the adsorption of intermediates, as well as accelerate phase transition of the OER process and decrease the adsorption of Cl^- [29,30]. As a result, $\text{Ru-Ni}_2\text{P}/\text{Ni}_5\text{P}_4$ achieve 1000 mA cm^{-2} with overpotentials of only 128 and 450 mV for HER and OER in alkaline seawater, respectively. After assembling the anion-exchange membrane water electrolyzer (AEMWE),

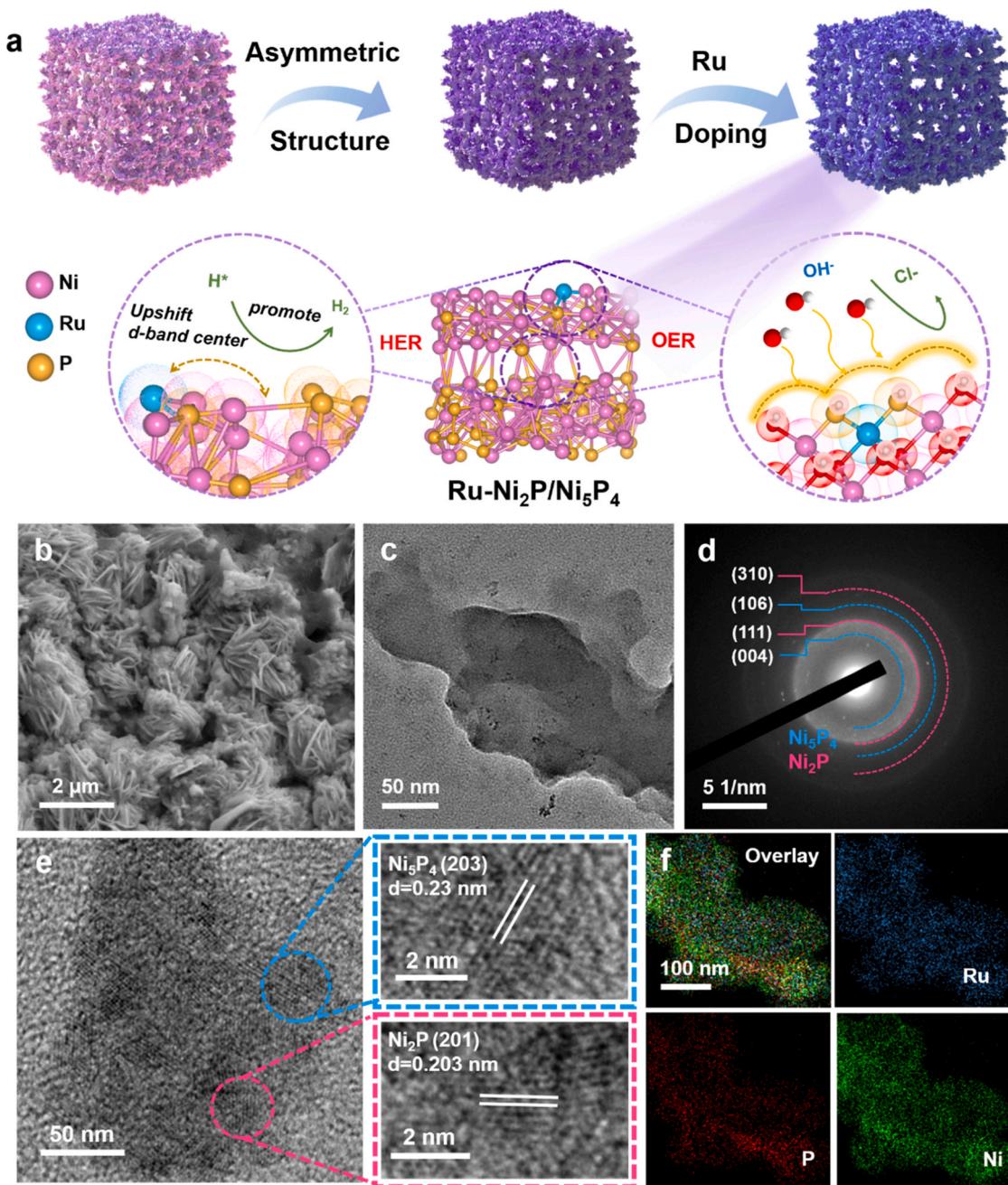


Fig. 1. (a) Schematic diagram of the preparation process for $\text{Ru-Ni}_2\text{P}/\text{Ni}_5\text{P}_4$. (b) SEM, (c) TEM, (d) typical SAED pattern, and (e) HRTEM images of $\text{Ru-Ni}_2\text{P}/\text{Ni}_5\text{P}_4$. (f) TEM mapping image of $\text{Ru-Ni}_2\text{P}/\text{Ni}_5\text{P}_4$.

Ru-Ni₂P/Ni₅P₄ exhibits an average efficiency of approximately 67% at different current densities, high energy intensity and low price of per GGE H₂ produced. The stability of the catalyst is demonstrated for more than 120 h at 500 mA cm⁻², proving its excellent resistance to Cl⁻ corrosion and showing its potential for industrial applications.

2. Result and discussion

2.1. Synthesis and characterization of Ru-Ni₂P/Ni₅P₄

Ru-doped Ni₅P₄/Ni₂P heterojunction structures were synthesized with abundant and dense nanosheet-like morphology through a fast molten salt and low-temperature phosphating strategy (Fig. 1a). The nickel foam (NF) with smooth surface and high electrical conductivity is used as support substrate (Fig. S1a). Firstly, the Ni₂(NO₃)₂(OH)₂·2 H₂O (NiNOOH) with uniform nanosheet-like structure is in-situ grown on NF by a fast molten salt strategy (Fig. S1b). Afterwards, the Ni₂P/Ni₅P₄ structure was synthesized by means of low-temperature phosphating, and the formation of nanoparticles could be observed by scanning

electron microscopy (SEM) (Fig. S1c), in which the formation of single or two phases of nickel phosphide could be realized by varying the phosphating time. Finally, doping of Ru elements in different concentrations of RuCl₃ solution (Ru-Ni₂P/Ni₅P₄-x, x=0.01, 0.05, 0.1), the formation of abundant and dense nanoparticles on nanosheets are observed (Fig. 1b). Transmission electron microscopy (TEM) images of Ru-Ni₂P/Ni₅P₄ further showed that the surface of Ru-Ni₂P/Ni₅P₄ is consisted of a large number of nanosheets with a homogeneous distribution (Fig. 1c). The formation of Ni₂P and Ni₅P₄ phases is further demonstrated by selective electron diffraction (SAED) pattern (Fig. 1d). Phosphating promotes the formation of the Ni_xP asymmetric structure and further high-resolution TEM (HRTEM) analysis reveal a spacing of 0.203 nm attributed to the (201) crystal plane Ni₂P (Fig. 1e), whereas a spacing of 0.23 nm is attributed to the (203) crystal plane Ni₅P₄, evidencing the formation of a heterogeneous interface. To further verify the composition and elemental distribution of the Ru-Ni₂P/Ni₅P₄ nanosheets, the energy-dispersive X-ray spectroscopy (EDX) elemental mappings analysis indicates that the elements of Ru, Ni, and P are uniformly distributed throughout the nanosheets without any obvious

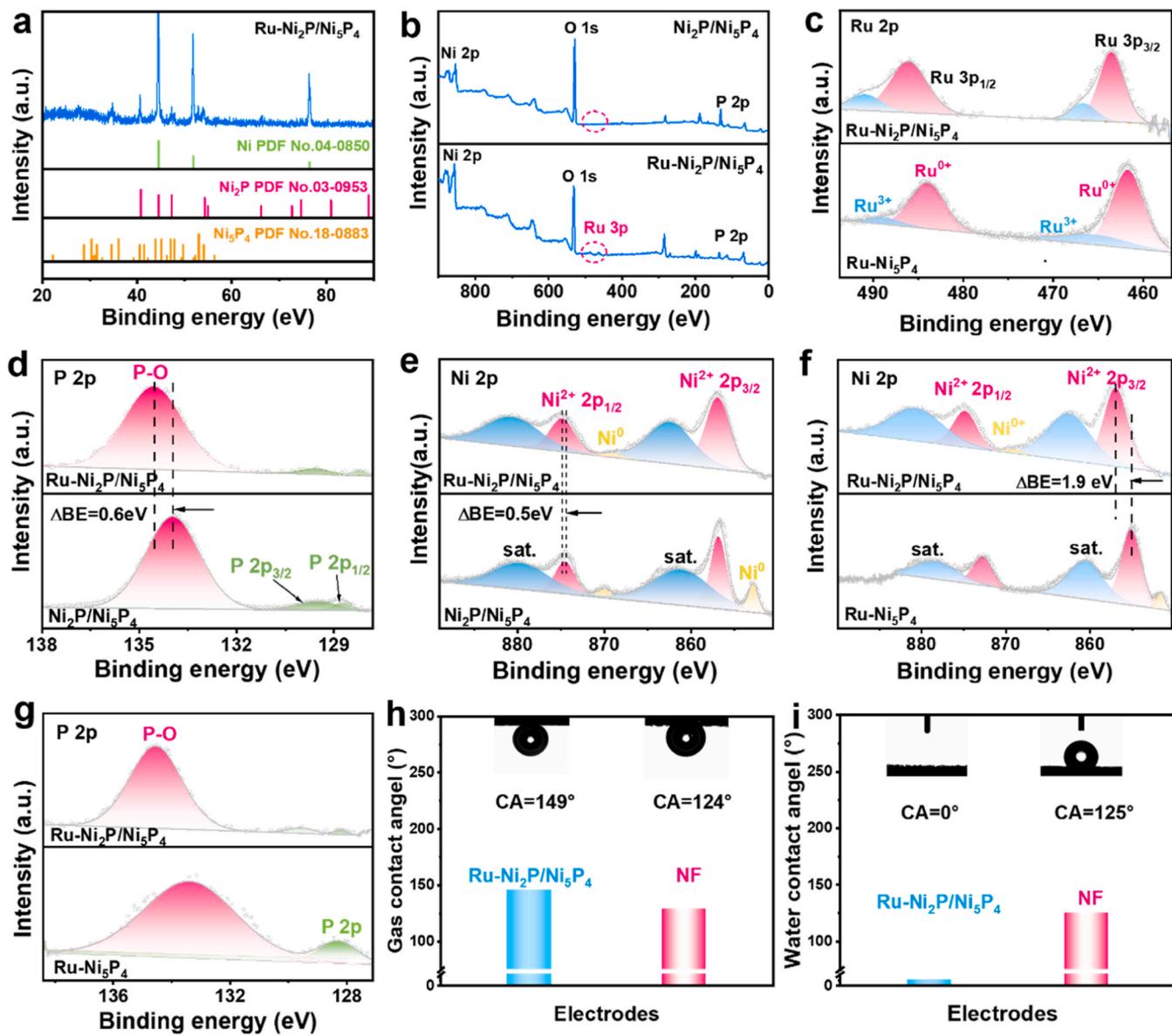


Fig. 2. (a) XRD pattern and (b) XPS survey spectra. High-resolution XPS spectra of (c) Ru 2p, (d, g) P 2p, (e, f) Ni 2p for as-prepared samples. (h, i) Photographs of contact angle measurements of Ru-Ni₂P/Ni₅P₄ and NF.

aggregation (Fig. 1f), which prove the homogeneous doping of Ru elements. The content of Ru in the nanosheets is determined using ICP-AES, and the content of Ru is only 2.28 wt% (Table S1), which is much lower than the weight percentage of commercial Pt/C and RuO₂ catalysts.

As shown in Fig. S2, the peaks of Ni₂P (JCPDS No.89-4864) and Ni₅P₄ (JCPDS No.84-2588) are shown in X-ray diffraction (XRD) after low-temperature phosphating prior to immersion in RuCl₃ solution, indicating the synthesis of the Ni₂P/Ni₅P₄ asymmetric structure. Fig. 2a confirms that XRD pattern of Ru-Ni₂P/Ni₅P₄ samples match well with the standard Ni₂P (JCPDS No.03-0953) and the standard Ni₅P₄ (JCPDS No.18-0883) and no phase of Ru monomers is observed, suggesting that the Ru element is in the doped form into the asymmetric structure. The XRD of the samples synthesized from molten salt corresponds to Ni (NO₃)₂(OH)₂ 2 H₂O (NiOOH) (JCPDS No. 00-027-0939) (Fig. S3). In order to investigate the electronic interactions of the Ru-Ni₂P/Ni₅P₄ catalyst, X-ray photoelectron spectroscopy (XPS) was performed. As can be seen from the comparative XPS spectra in Fig. 2b, Ru element is successfully introduced in Ru-Ni₂P/Ni₅P₄, both Ni₂P/Ni₅P₄ and Ru-Ni₂P/Ni₅P₄ catalysts indicate the existence of Ru, Ni and P elements. As shown in Fig. 2c, peaks at 486 and 463.4 eV in the Ru 3p spectrum of Ru-Ni₂P/Ni₅P₄ are attributed to the Ru 3p_{3/2} and Ru 3p_{1/2} of Ru⁰, and the peaks at 490.8 and 466.4 eV are two peaks of Ru³⁺ [31,32]. Moreover, comparing the peaks of Ru 2p of Ru-Ni₂P/Ni₅P₄ with Ru-Ni₅P₄, it can be seen that Ru 2p of Ru-Ni₂P/Ni₅P₄ is shifted towards higher binding energy, suggesting that more electrons are transferred towards the asymmetric structure interface. For Fig. 2d, the fitted peaks corresponding to 134.5 and 133.9 eV indicate the presence of P-O bonds in Ru-Ni₂P/Ni₅P₄ and Ni₂P/Ni₅P₄, and the other two peaks located near 129.5 eV are attributable to the 2p_{1/2} and 2p_{3/2} of M-P [33,34]. Fig. 2e shows the high-resolution XPS spectra of Ni can be divided into five main peaks. Peaks at 874.8 and 862.5 eV belong to the 2p_{1/2} and 2p_{3/2} orbitals of Ni²⁺, peaks at 880.7 and 869.1 eV belong to the satellite peaks, and peaks at 869.1 eV belongs to Ni⁰ [35]. Therefore, comparing the XPS of the two catalysts for Ni 2p and P 2p spectra of both Ni 2p and P 2p orbitals, the doping of Ru result in an overall shift of the peaks of both Ni 2p and P 2p orbitals towards higher binding energies, suggesting that the doping of Ru change the electronic environment at the Ni₂P/Ni₅P₄ asymmetric heterointerface [36]. To further explore the effect of constructing asymmetric structure on the catalyst, the XPS of Ni 2p and P 2p of Ru-Ni₂P/Ni₅P₄ and Ru-Ni₅P₄ are compared (Figs. 2f and 2g) and Ru-Ni₂P/Ni₅P₄ with asymmetric structure shift towards higher binding energies, suggesting that Ru doping in Ni_xP asymmetric structure has a more pronounced electron-transfer process. In order to demonstrate the superiority of the Ru-Ni₂P/Ni₅P₄ catalyst, contact angle tests (CA) were performed. Firstly, the contact angle test with water is compared, and when the water droplets are contacted with the Ru-Ni₂P/Ni₅P₄, they are rapidly penetrated with a contact angle close to 0°, proving the super hydrophilicity of the catalyst. By contrast, the smooth surface NF without catalyst nanosheets loading show poorer hydrophilicity with a CA of 125° (Fig. 2h). Moreover, gas bubble CA tests indicate that Ru-Ni₂P/Ni₅P₄ and pure NF have CA of 149° and 124°, respectively, and Ru-Ni₂P/Ni₅P₄ exhibits superior gas transport performance (Fig. 2i). The unique nanosheets morphology of Ru-Ni₂P/Ni₅P₄ contributes to surface wettability and bubble release during water electrolysis [37]. Thus, the hydrophilic gas-transport characteristic of Ru-Ni₂P/Ni₅P₄ nanosheets can accelerate water dissociation and bubbles diffusion, which is conducive to the seawater electrolysis.

2.2. Electrocatalytic performance of Ru-Ni₂P/Ni₅P₄ investigation for HER and OER

The catalytic activities of Ru-Ni₂P/Ni₅P_{4-x} (x=0.01, 0.05, 0.1) with different Ru loading was firstly investigated in a typical three-electrode system (Fig. S4a and S4b). By comparing the catalysts with different Ru contents, it can be seen that the Ru content of 5 mg mL⁻¹ concentration is the optimal choice after the comprehensive performance and

economic value considerations. Based on that, The Ru content of 5 mg mL⁻¹ was selected for subsequent tests. In order to investigate the effects of asymmetric Ni_xP structure as well as Ru doping on the catalyst performance, LSV curves of Ru-Ni₂P/Ni₅P₄, Ru-Ni₅P₄ and commercial Pt/C and RuO₂ performances were compared under the same conditions (Fig. 3a). Apparently, Ru-Ni₂P/Ni₅P₄ exhibits excellent HER and OER performance, and the LSV curves show that low overpotentials of only 128 and 450 mV are required to achieve high current density at 1000 mA cm⁻² for HER and OER, respectively, much smaller than the overpotentials of Ru-Ni₅P₄ (200 and 604 mV), the benchmark commercial Pt/C (227 mV) and RuO₂ (470 mV) (Fig. 3b). The Tafel slopes of different catalysts show that the Ru-Ni₂P/Ni₅P₄ nanosheets exhibit the smallest Tafel slopes in HER and OER (52 and 51 mV dec⁻¹), respectively, which are even lower than those of commercial Pt/C and RuO₂ (71 and 74 mV dec⁻¹), indicating that the Ru-Ni₂P/Ni₅P₄ electrode display fast HER and OER kinetics (Fig. S5a and S5b). Based on electrochemical impedance spectroscopy (EIS) measurements, the Ru-Ni₂P/Ni₅P₄ electrode exhibit the smallest semicircle and the lowest charge transfer resistance (R_{ct}) values in the low frequency range compared with the control sample (Fig. S6a and S6b), suggesting a fast charge transfer resistance within the Ru-Ni₂P/Ni₅P₄ catalyst, which is consistent with the LSV curves and Tafel slopes. The Ru-Ni₂P/Ni₅P₄ nanosheets have excellent activity, which is well ahead of the recently reported bifunctional HER and OER catalysts (Fig. 3c and Table S2). In addition, we obtain cyclic voltammogram curves with different scan rates at non-Faraday region (Fig. S7-S8), and then fit the double layer capacitance (C_{dl}), as shown in Fig. S9. It can be found that the Ru-Ni₂P/Ni₅P₄ exhibits the largest C_{dl} value, which further reflects that the Ru-Ni₂P/Ni₅P₄ can provide more active sites. Fig. 3d summarize the combined comparison of the HER and OER performance of Ru-Ni₂P/Ni₅P₄ with that of Ni₂P/Ni₅P₄ and Ru-Ni₅P₄ based on the above mentioned tests, suggesting that Ni_xP asymmetric structure with Ru doping improve the HER and OER performance.

Moreover, the HER and OER performance of the catalysts in different alkaline solutions were tested in simulated seawater (1.0 M KOH + 0.5 M NaCl) and real seawater (1.0 M KOH + seawater). As shown in Fig. S10-S15, the LSV, Tafel slope, EIS and C_{dl} performance is compared in simulated seawater and real seawater, respectively, and find that its performance in seawater is not much different from that in KOH, thus proving the corrosion resistance to Cl⁻ during seawater electrolysis.

It is well known that tertiary butyl alcohol (TBA) is a special hydrogen radical eradicator, which is an effective method to study the strength of hydrogen adsorption. As shown in Fig. S16, the HER performance of Ru-Ni₂P/Ni₅P₄ and Ru-Ni₅P₄ is significantly decreased by the addition of 1.0 M TBA, in which the current density of Ru-Ni₅P₄ decrease more significantly, and the overpotentials of Ru-Ni₂P/Ni₅P₄ at 400, 600, and 800 mA cm⁻² are 2.54, 2.21, and 2.04 times higher than those of Ru-Ni₅P₄, which means that Ru-Ni₅P₄ is more susceptible to the TBA bursting agent, indicating that the asymmetric structure surface has more H* as well as stronger hydrogen adsorption strength (Fig. 3e), which is beneficial to improve the HER process [38,39]. The charge transfer kinetics of the inner layer of the electrode material can be reflected by monitoring the charge transfer resistance (R_{ct}) and double layer capacitance (CPE₁) in the first parallel assembly. A second parallel circuit was preferred to be analysed to truly reflect the charge relaxation behavior of the HER intermediates (Figure S17). Concretely, the adsorption behavior of hydrogen intermediates on chemically active sites can be reflected by fitting the R_{ct} and hydrogen adsorption pseudo capacitance (C_{ph}) (Fig. 3f). It is not difficult to see that the hydrogen adsorption charge (Q_{H^*}) calculated by integrating the processed C_{ph} is greatly improved for Ru-Ni₂P/Ni₅P₄ compared with the Q_{H^*} of Ru-Ni₅P₄, which is sufficient to prove that the Ni_xP asymmetric structure greatly improves the H* adsorption strength on Ru-Ni₂P/Ni₅P₄ [40]. In order to demonstrate that Ru-Ni₂P/Ni₅P₄ has a better OH* adsorption capacity than Ru-Ni₅P₄, the tests were carried out in KOH-H₂O and KOH-D₂O electrolytes respectively. The presence of a stronger deuterium bonding

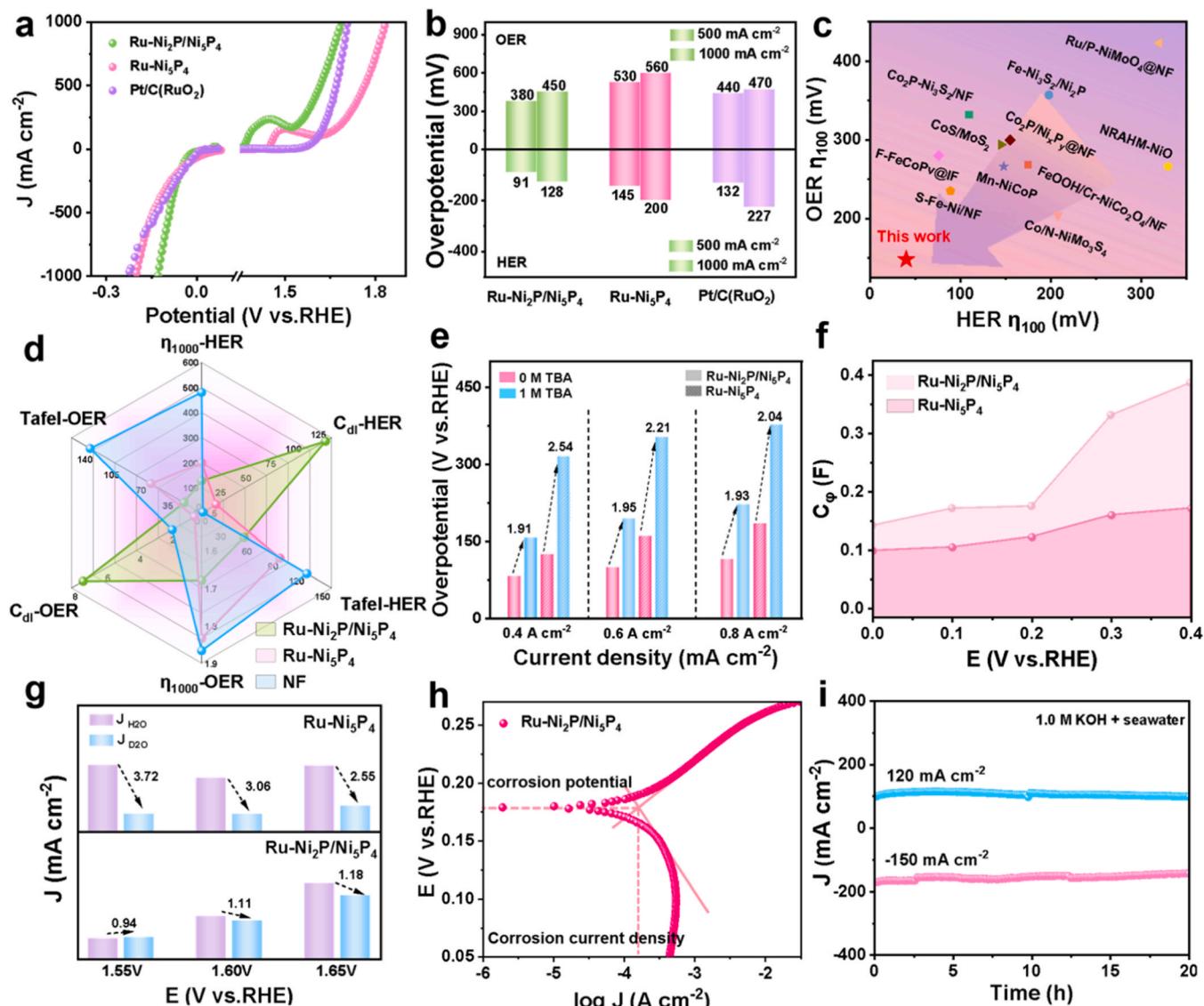


Fig. 3. (a) LSV performance of different samples and (b) corresponding HER and OER overpotentials at different current density. (c) Comparison of the overpotential for HER and OER with other electrocatalysts. (d) Comprehensive comparisons of the HER and OER performance. (e) The overpotentials at different current density of Ru-Ni₂P/Ni₅P₄ with other samples in 1.0 M KOH with and without TBA. (f) Plots of C_φ vs η during HER, (g) OER kinetics investigations, and (h) Tafel plots of Ru-Ni₂P/Ni₅P₄ in 1.0 M KOH + seawater. (i) Chronoamperometry i-t curve of Ru-Ni₂P/Ni₅P₄ for HER and OER.

network in D₂O than the O-H in H₂O, and the fact that the O-D bonds in D₂O are more difficult to break during catalysis due to their stronger bonding energy than the O-H bonds. Therefore, the OER overpotential of Ru-Ni₂P/Ni₅P₄ occurring in KOH-D₂O increases relative to that of KOH-H₂O (Fig. S18). In addition, it was proved by calculation that Ru-Ni₂P/Ni₅P₄ has stronger adsorption of OH[·]. Therefore, in D₂O, the smaller the decay, the stronger the adsorption of OH[·] is proved and the smaller the overpotential is required (Fig. 3g) [41]. In particular, due to the formation of ClO[·] corrosion by Cl[·] oxidation during the OER process, we test two samples for their resistance to Cl[·] corrosion [42], and it is clear that Ru-Ni₂P/Ni₅P₄ has higher corrosion potential, which indicates its excellent corrosion resistance (Fig. 3h and S19).

In considering the practical application of catalysts, durability is an important factor. As shown in Fig. 3i, in alkaline seawater, the Ru-Ni₂P/Ni₅P₄ catalyst shows negligible current attenuation after sustained electrolysis for 20 h at about 120 and 150 mA cm⁻², respectively, suggesting that the Ru-Ni₂P/Ni₅P₄ has excellent stability. After stability testing, TEM and XPS of Ru-Ni₂P/Ni₅P₄ were further determined. As shown in Fig. S20, it can be clearly observed that the catalyst maintains

the nanosheet-like structure after HER and the formation of the amorphous layer can be clearly observed in TEM after the stabilization of OER due to the anodic oxidation reaction. From the XPS spectra of Ni 2p (Fig. S21), it can be observed that a new peak appeared at 865.1 and 881.4 eV belong to the peak of Ni³⁺, indicating the generation of NiOOH during the OER process. And the peak of PO₄³⁻ appearing in the P 2p spectrum carries an anionic charge that has a homogeneous repulsion with Cl[·], thus greatly avoiding the corrosion of the catalyst. As shown in Fig. S22, the nearly unchanged peaks of the Ni 2p, P 2p, and Ru 2p spectra in the XPS after HER demonstrates that no phase transition occurs during the HER process. In addition, we further detected the generation of ClO[·] in the electrolyte after the OER reaction by a colourimetric method. When free chlorine is present, N, N-diethyl-p-phenylenediamine sulphate reacts to form a red dye with strong absorption. Therefore, it was measured by UV-vis spectrophotometer (Fig. S23). As shown in Fig. S24, no ClO[·] was generated after the OER reaction, indicating that no chlorine gas was produced as a by-product [43].

2.3. Mechanism of improving HER and OER performance of Ru-Ni₂P/Ni₅P₄

For the possible changes in the HER and OER processes, we tested the possible phase transitions in the HER and OER processes by in-situ Raman. Fig. 4a demonstrates the Raman comparison of Ru-Ni₂P/Ni₅P₄ and Ru-Ni₅P₄ in the HER voltage range, respectively, and neither of them shows the generation of new phases in the HER voltage range, and only Ni-P bonds can be indicated, which proves that the phase transition does not occur in the HER process. For the OER process, since the transition metal catalyst undergoes an irreversible phase transition process during the OER process, and thus the oxyhydroxide generated on the surface of the catalyst are considered to be the real active sites in the OER process. Therefore, the in-situ Raman of Ru-Ni₂P/Ni₅P₄ and Ru-Ni₅P₄ under OER conditions are shown in Figs. 4b and 4c, respectively, and it can be clearly observed that the peaks of NiOOH in Ru-Ni₂P/Ni₅P₄ appear at a voltage of 1.45 V, whereas those of NiOOH in Ru-Ni₅P₄ appear at 1.55 V, indicating that the construction of Ni₂P/Ni₅P₄

asymmetric structure in the OER process could accelerate the formation of NiOOH as the OER active sites [44,45]. Subsequently, the theoretical model of Ru-Ni₂P/Ni₅P₄, Ru-Ni₅P₄, Ru/P-NiOOH were constructed (Fig. S25). As shown in Fig. S26, when Ru atom is doped, it causes a change in the charge density around the Ni and P atoms near the Ru atom. In addition, we also compare the charge changes when the Ru element is doped in Ni₅P₄ monophase and Ni₂P/Ni₅P₄ asymmetric structure. Undoubtedly, Ru transfers more electrons (0.079 e) to the Ni_xP asymmetric structure, which contributes to the excitation of Ni_xP activity. The total density of states (TDOS) of Ru-Ni₂P/Ni₅P₄ and Ru-Ni₅P₄ were further calculated (Fig. 4d), and the comparison reveal that Ru-Ni₂P/Ni₅P₄ has the highest charge density near the Fermi energy level, suggesting an enhanced conductivity. Moreover, comparing the d-band center of Ru-Ni₅P₄ (-1.23 V), the d-band center of Ru-Ni₂P/Ni₅P₄ (-1.19 V) is closer to the Fermi energy level, suggesting that the Ru doping and Ni₂P/Ni₅P₄ asymmetric structure facilitate the adsorption of intermediates during the reaction process and enhances the electrochemical process. Furthermore, Ru-Ni₂P/Ni₅P₄ has the lowest work

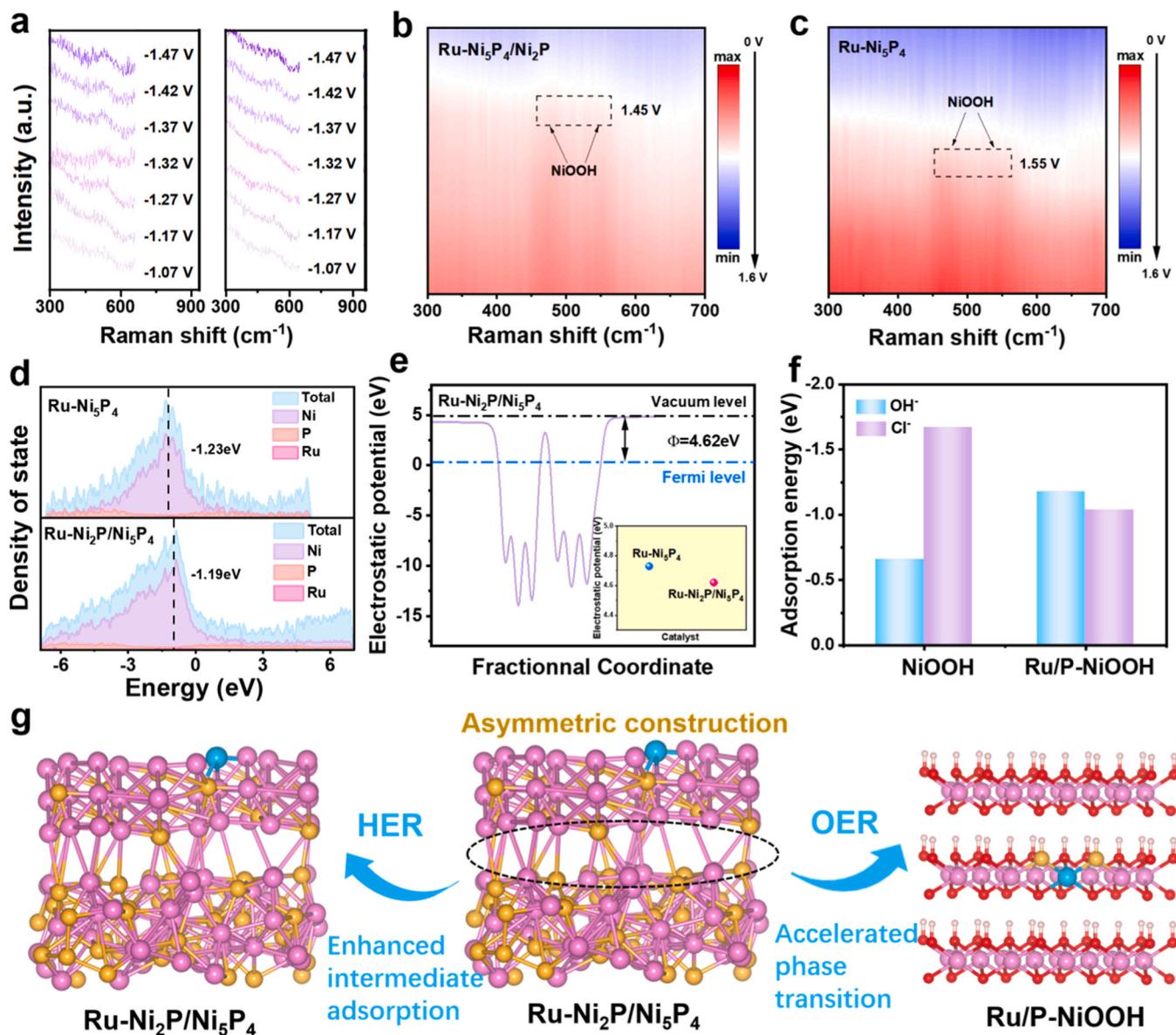


Fig. 4. In-situ Raman for (a) Ru-Ni₂P/Ni₅P₄ and Ru-Ni₅P₄ in HER. In-situ Raman of (b) Ru-Ni₂P/Ni₅P₄ and (c) Ru-Ni₅P₄ in OER. (d) DOS of Ru-Ni₂P/Ni₅P₄, Ru-Ni₅P₄ and Ni₂P/Ni₅P₄. (e) The computed work functions of Ru-Ni₂P/Ni₅P₄. (f) Comparison of Cl⁻ and OH⁻ adsorption energies for NiOOH and Ru/P-NiOOH. (g) Schematic representation of the genuine phases and active sites of Ru-Ni₂P/Ni₅P₄ in 1.0 M KOH seawater during HER and OER.

function (4.62 eV), which is lower than that of Ru-Ni₂P₄ (5.14 eV) (Fig. 4e and S27), suggesting that electrons are more likely to spill over from the interior of Ru-Ni₂P/Ni₅P₄ to the surface and exchange electrons with the reactants, thus better accelerating the electrocatalytic dynamic process. In addition, the adsorption energy of OH⁻ from seawater by Ru/P-NiOOH in the OER process is higher than that of Cl⁻, thus demonstrating the excellent Cl⁻ repellency in the OER process (Fig. 4f). In conclusion, the introduction of Ni₂P/Ni₅P₄ asymmetric structure in Ru-Ni₂P/Ni₅P₄ promotes the adsorption of intermediates in HER process, also facilitates the electron transfer and fast phase transition in OER process, and repels Cl⁻ simultaneously, which is conducive to the efficient and stable overall seawater electrolysis (Fig. 4g).

2.4. Investigation of electrocatalytic activity for overall seawater splitting

Based on the revelation of the excellent bifunctional catalytic activity of Ru-Ni₂P/Ni₅P₄ catalysts in alkaline seawater, the activity of Ru-Ni₂P/Ni₅P₄ in a two-electrode electrolyzer by simulating industrial electrolysis conditions in 1.0 M KOH + seawater solutions was tested and compared it with commercial Pt/C and RuO₂. As shown in Fig. 5a, it can be observed that the comparison of the LSV curves reveals that the performance of Ru-Ni₂P/Ni₅P₄ couple in seawater is much better than that of commercial Pt/C and RuO₂. In 1.0 M KOH + seawater, the voltage required for a large current density of 1000 mA cm⁻² is only 1.83 V, which is much smaller than that of commercial Pt/C and RuO₂ (2.09 V). Significantly, the performance of Ru-Ni₂P/Ni₅P₄||Ru-Ni₂P/Ni₅P₄ barely exceeds that of the most recorded catalysts (Fig. 5b, Tables S3 and S4). In addition, when the chronoamperometry i-t curve is measured in 1.0 M KOH + seawater, it can last for 100 h at a current density of 100 mA cm⁻² almost no degradation (Fig. 5c). Subsequently, calculated Faraday efficiency by collecting hydrogen from the cathode and oxygen from the anode in 1.0 M KOH + seawater based on the drainage method, where bubbles were produced at both the cathode and anode of the electrolyzer, which could be reflected by the decrease of the

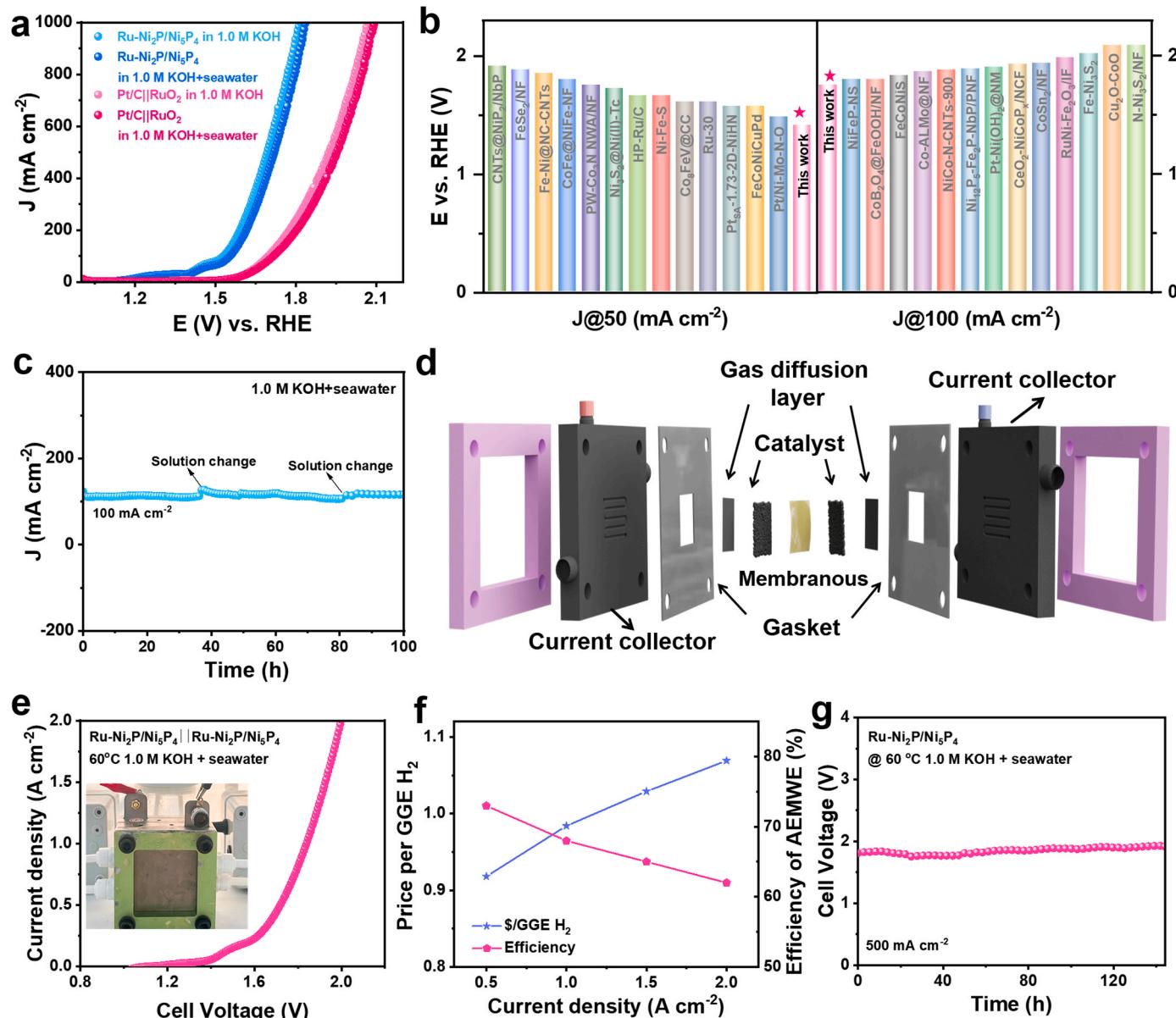


Fig. 5. (a) LSV curves in different solutions for overall seawater splitting. (b) Comparison of operation voltages required to achieve 50 and 100 mA cm⁻² with benchmarking works. (c) Chronoamperometry curve of Ru-Ni₂P/Ni₅P₄ in 1.0 M KOH seawater. (d) Schematic diagram of the AEMWE electrolyzer. (e) Polarization curves of Ru-Ni₂P/Ni₅P₄||Ru-Ni₂P/Ni₅P₄ electrodes, the inset is a picture of the AEMWE device. (f) Efficiency of AEMWE electrolyzer at different current densities and price of per GGE H₂. (g) Chronopotentiometry curve of Ru-Ni₂P/Ni₅P₄ at 500 mA cm⁻².

solution in the test tube. As shown in Fig. S28, the obtained V_{H2}/V_{O2} value is calculated to be 2:1, thus inferring that the catalyst has a Faraday efficiency close to 100% in 1.0 M KOH + seawater. The overall seawater electrolysis process can then be driven by a stirling engine (Fig. S29a), and bubble generation on the catalyst surface can be clearly observed. In addition, HER and OER bifunctional performance of the Ru-Ni₂P/Ni₅P₄ electrocatalyst can be demonstrated based on the energy supplied by the solar panel (Fig. S29b), which can be accomplished with only 1.82 V to achieve 130 mA cm⁻², further proving that the Ru-Ni₂P/Ni₅P₄ catalyst has obvious potential for seawater electrolysis. In order to verify its potential for industrial application, Ru-Ni₂P/Ni₅P₄ was assembled as cathode and anode in an AEMWE to test its performance. Fig. 5d shows the schematic of the AEMWE device, and we subsequently simulate the industrial conditions of 60 °C seawater for testing, as shown in Fig. 5e, where a high current density of 1000 mA cm⁻² can be achieved at a voltage of only 1.83 V. In addition, we evaluate the economic viability of the AEM electrolyzer with an average efficiency of AEMWE is approximately 67% at different current densities and a price of per gasoline-gallon equivalent (GGE) H₂ of \$1 [42], far below the U.S. Department of Energy's price standard of \$2 in 2026 (Fig. 5f). In order to verify the stability of the Ru-Ni₂P/Ni₅P₄ catalysts in AEMWE, long-term stability tests were subsequently performed under simulated industrial conditions. As shown in Fig. 5g, the Ru-Ni₂P/Ni₅P₄ catalyst is able to operate stably at a current density of 500 mA cm⁻² for more than 120 h without significant voltage drop.

3. Conclusion

In this work, bifunctional Ru-doped Ni_xP asymmetric electrocatalysts (Ru-Ni₂P/Ni₅P₄) for alkaline seawater-based anionic membrane (AEM) electrolyzer has been prepared. In situ characterizations and theoretical calculations show that the introduction of asymmetric structure manipulate the electronic structure and d-band center of Ru-Ni₂P/Ni₅P₄ and optimize the adsorption of the active site for H^{*}, resulting in an enhanced HER activity. Moreover, the introduction of the asymmetric structure lead to a rapid phase transition of Ru-Ni₂P/Ni₅P₄ into real oxyhydroxide active species. The co-doping of Ru/P elements modulate the adsorption energies of Cl⁻ and OH⁻, enhancing the selectivity of OH⁻ by suppressing the erosion of Cl⁻, thus ensuring the long cycle stability of the Ru-Ni₂P/Ni₅P₄ for seawater electrolysis. The Ru-Ni₂P/Ni₅P₄ integrated as both cathode and anode in an alkaline seawater AEMWE is able to operate continuously for more than 100 h at a high current density of 500 mA cm⁻² and show great economics. This work has led to the design of asymmetric electrocatalyst with high activity, stability and selectivity for industrial seawater electrolysis.

CRediT authorship contribution statement

Yanan Xia: Investigation, Data curation, Conceptualization, Formal analysis, Validation, Writing-original draft. Conceptualization, Writing-review & editing, Supervision. **Lili Guo:** Formal analysis, Data curation, Conceptualization, Validation. **Jiawei Zhu:** Supervision, Formal analysis, Data curation. **Junheng Tang:** Data curation, Conceptualization, Validation. **Zhipeng Li:** Conceptualization, Validation. **Xiaobin Liu:** Supervision, Validation, Conceptualization. **Jingqi Chi:** Supervision, Validation, Funding acquisition. **Lei Wang:** Writing-review & editing, Funding acquisition, Supervision.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data Availability

The data that has been used is confidential.

Acknowledgements

This work is financially supported from the National Natural Science Foundation of China (52072197, 52174283, and 22301156), the Natural Science Foundation of Shandong Province (ZR2021QE165), Youth Innovation and Technology Foundation of Shandong Higher Education Institutions, China (2019KJC004), Major Scientific and Technological Innovation Project (2019JZZY020405), Major Basic Research Program of Natural Science Foundation of Shandong Province under Grant (ZR2020ZD09), Shandong Province "Double-Hundred Talent Plan" (WST2020003), Taishan Scholar Young Talent Program (tsqn201909114), and University Youth Innovation Team of Shandong Province (202201010318).

Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at doi:10.1016/j.apcatb.2024.123995.

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